

Evaluate Morphological Effect on MR Relaxivity for Bifunctional Au-Fe₃O₄ Heterostructures

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Introduction: Au-Fe₃O₄ heterostructures are one of the typical magnetic-based nanocomposites that are usually used in biomedical applications, such as dual functional probes for optical imaging and magnetic resonance imaging¹. The heterostructures have been designed into various architecture, including core-shell, hetero-oligomer, and core-satellite for using as MRI contrast agents. Previous studies have proved that different composites of magnetic-based nanoparticles (NPs) may reveal diverse characteristics while being used as MRI contrast agents^{2,3}. Although Au-Fe₃O₄ heterostructures may exhibit weaker T₂-decaying effect than pure Fe₃O₄ NPs because of the shielding effect of Au NPs, it is still worthy to investigate Au-Fe₃O₄ heterostructures for their multifunctional characteristics of diagnostic and therapeutic applications. In these decades several magnetite-based heterostructures have been developed for multi-modal imaging probes. However, the effect of morphology of different composite heterostructures on MRI relaxivity is rarely reported. In this study, pure Fe₃O₄ NPs, 5 nm Au dumbbell-like NPs (mAu DBNPs), 10 nm Au dumbbell-like NPs (LAu DBNPs), and 10 nm Au flower-like NPs (LAu FLNPs) were synthesized to demonstrate the T₂-decaying effect among different morphological iron-based heterostructures.

Methods: The Fe₃O₄ NPs and Au-Fe₃O₄ heterostructures were synthesized according to previous work⁴. The heterostructured Au-Fe₃O₄ NPs were prepared by thermal decomposition of iron-oleate complex (Fe(OL)₃) in the presence of different sizes of Au NPs, saying 5 nm and 10 nm in this study. Dumbbell-like NPs (DBNPs) and flower-like NPs (FLNPs) were manufactured by adding different concentrations of (Fe(OL)₃). It has been reported that once the Fe₃O₄ starts to nucleate on an Au NPs, the free electrons from the Au must compensate for the charge induced by the polarized plane at the interface⁵. Since the 5 nm Au NPs have only limited number of electrons, this compensating effect makes all other facets of the Au NPs electron deficient and leads to be unsuitable for multi-nucleation. Consequently, only the dumbbell-like structure of Au-Fe₃O₄ can be composed. As for 10 nm Au NPs, the larger Au provides large surface areas and multiple monocrystalline domains for Fe₃O₄ to nucleation, resulting in the production of dumbbell-like or flower-like heterostructures, depending on the amounts of adding (Fe(OL)₃). The dimension and morphology of Fe₃O₄ and Au-Fe₃O₄ heterostructures were examined by analytical transmission electron microscopy (TEM). The concentrations of Fe and Au atom were quantitatively measured by inductively coupled plasma atomic emission spectrometry (ICP-AES) (Kontron, S-35). All MR acquisitions were performed with multi-echo spin echo pulse sequence in a 7 Tesla animal MRI scanner (ClinScan 70/30 USR, Bruker, Germany) with following parameters: TR = 2000 ms, TE = 5.7 to 182.4 ms with an interval of 5.7 ms, number of echoes = 32, voxel size = 0.156 mm × 0.156 mm × 2.0 mm. Seven trials were performed with five concentrations of iron between 0.07–0.42 mM. Various concentrations of the Au-Fe₃O₄ heterostructures were dispersed in bidistilled water. T₂ values and r₂ relaxivity were evaluated to comprehend the T₂-decaying ability of Au-Fe₃O₄ heterostructures with various morphologies and structures. The signal intensities within manually determined regions of interest (ROI) were analyzed by home-developed program with usage of MATLAB (The Mathworks) software. To verify reproducibility, the same samples were performed twice.

Results: Figure 1 showed TEM images of pure Fe₃O₄ (A) and different Au-Fe₃O₄ heterostructures of mAu DBNPs (B), LAu DBNPs (C) and LAu FLNPs (D). The mean diameters of Fe₃O₄, mAu DBNP, LAu DBNP, and LAu FLNP were 13, 13±1, 14±1, and 24±2 nm, respectively. The enlarged insertions demonstrated the appearance of dumbbell-like and flower-like NPs. The T₂-decaying effect on Fe₃O₄ heterostructures at various Fe concentrations was shown by T₂WI in Fig.2. The signal intensities decreased with increased Fe concentrations. Fe₃O₄ NPs displayed the most strong T₂-decaying effect while LAu DBNPs displayed relative weaker T₂-decaying ability. Figure 3 displayed the results of R₂ (1/T₂) of one of the seven trials. A good linearity between R₂ and Fe concentrations was observed in each sample, reflected by high correlation coefficients (R²) of 0.93–1.00. To further compare the T₂-decaying ability of each composite, the values of r₂ relaxivity, i.e. the slope of fitting curve shown in Fig. 3, were computed. As shown in Fig. 4, the r₂ relaxivities were 130.1±16.3, 114.8±20.6, 82.2±31.8 and 104.3±30.8 mM⁻¹s⁻¹ for Fe₃O₄, mAu DBNPs, LAu DBNPs, and LAu FLNPs, respectively. The repeated experiment showed the % differences of 1.7%, 4.1%, 6.4%, and 8.2% for Fe₃O₄, mAu DBNPs, LAu DBNPs, and LAu FLNPs, respectively, showing the high reproducibility of measured r₂ relaxivities.

Discussion and Conclusions: In this study, the T₂-decaying ability of various Au-Fe₃O₄ heterostructures with different sizes of Au and with different morphologies of Au-Fe₃O₄ heterostructures was investigated. For 5 nm Au seeds, only dumbbell-like NPs could be produced. In contrast, using 10 nm Au seeds could produce both of dumbbell-like and flower-like NPs. Considering the size of Au, heterostructures with smaller Au NPs have less influence on the magnetic behavior, which attributed to the less contact area and weaker junction effect between smaller Au NPs and Fe₃O₄ NPs¹. Therefore, mAu

DBNPs showed stronger r₂ relaxivity than that of LAu DBNPs, which is consistent with a previous report¹. In addition, the impact of different morphological Au-Fe₃O₄ heterostructures on MRI relaxivity was also investigated in this study. It was found that LAu FLNPs exhibited higher relaxivity than that of LAu DBNPs. This improvement might result from the synergistic effect which more than one iron oxide NPs were linked to one single Au NP, and therefore resulted in magnetic coupling effect among Fe₃O₄ NPs⁶. Accordingly, the magnetic homogeneity surrounding water molecules were deteriorated by multiple tiny magnets around their vicinity. Moreover, the competing forces of inter-particle and intra-particle interactions were shown among FLNPs while DBNPs only have inter-particle interaction. Consequently, the FLNPs revealed stronger T₂-decaying effect. In conclusion, using different morphological Au-Fe₃O₄ heterostructures or different sizes of Au NPs as seeds could result in different r₂ relaxivity. To manufacture proper configurations of heterostructures are able to enhance T₂-decaying effect and can improve image contrast of T₂ weighted images in MRI applications.

References: (1) Xu C. et al. Chem. Int. Ed. 2008, 47, 173–176. (2) Zhou, T. et al. J Mater Chem 2012, 22, 470–477. (3) Garcia, I. et al. Bioconjugate Chem 2011, 22, 264–273. (4) Lin, F. H. et al. Nano Res 2011, 4, 1223–1232. (5) Heng Y. et al. Nano Lett., 2005, 5, 379–382. (6) Berret, J. F. et al. J Am Chem Soc 2006, 128, 1755–1761.

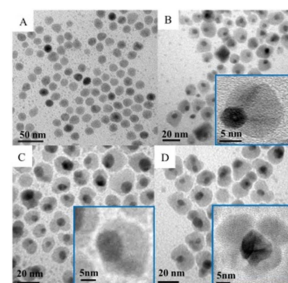


Fig. 1. TEM images of pure Fe₃O₄ (A) and Au-Fe₃O₄ heterostructures of mAu DBNPs (B), LAu DBNPs (C) and LAu FLNPs (D).

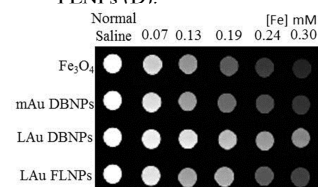


Fig. 2. T₂WI of Fe₃O₄ and Au-Fe₃O₄ heterostructures at various Fe concentrations. Fe₃O₄ NPs displayed the most strong T₂-decaying effect while LAu DBNPs displayed relative weaker T₂-decaying ability.

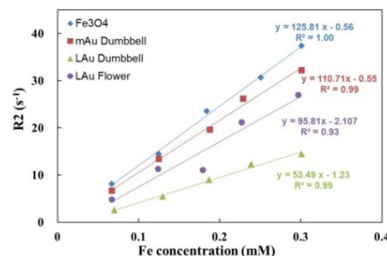


Fig. 3. High linearity was shown between R₂ and Fe concentrations, reflected by high R² of 0.93–1.00.

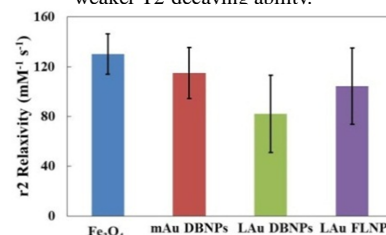


Fig. 4. The r₂ relaxivities were 130.1±16.3, 114.8±20.6, 82.2±31.8 and 104.3±30.8 mM⁻¹s⁻¹ for Fe₃O₄ and Au-Fe₃O₄ heterostructures.